

Synthesis of mesoporous aluminophosphates using surfactants with long alkyl chain lengths and triisopropylbenzene as a solubilizing agent

Tatsuo Kimura,^a Yoshiyuki Sugahara^a and Kazuyuki Kuroda^{*a,b}

^a Department of Applied Chemistry, School of Science and Engineering, Waseda University, Ohkubo-3, Shinjuku-ku, Tokyo 169, Japan

^b Kagami Memorial Laboratory for Materials Science and Technology, Waseda University, Nishiwaseda-2, Shinjuku-ku, Tokyo 169, Japan

Mesostructured organo-aluminophosphates were prepared by using surfactants with long alkyl chain lengths (C₁₆ and C₂₂) and by utilizing solubilization of triisopropylbenzene; calcination of the mesostructured materials yielded thermally stable mesoporous aluminophosphates with surface areas >700 m² g⁻¹ and with pore diameters in the range 1.8–3.9 nm.

Mesoporous silicas, denoted FSM¹ and M41S,² are very attractive materials for catalysts, catalyst supports and adsorbents for larger molecules than those treated in microporous crystals. Since microporous AlPO_{4-n} materials are known,³ it is expected that mesostructured organo-aluminophosphates are useful for the formation of mesoporous aluminophosphates to enlarge the variety and possible applications of mesoporous materials. We have already reported the successful formation of a hexagonal mesostructured aluminophosphate (AIPO) by using hexadecyltrimethylammonium chloride.⁴ Feng *et al.* also reported a hexagonal mesostructured material prepared in the presence of F⁻ ions, however, the porous structure collapsed upon calcination.⁵ Recently a mesoporous aluminophosphate with large surface area (790 m² g⁻¹) was reported by Zhao *et al.*⁶ Although the pore diameter of the product was reported to be *ca.* 4.0 nm, this value is somewhat questionable because it is too large judging from the values observed for ordered mesoporous silicas.^{1,2} On the other hand, Holland *et al.*⁷ and Cheng *et al.*⁸ reported mesoporous aluminophosphate-based materials *via* layered intermediates; the porous materials, prepared by solvent extraction, are thermally unstable.

The hexagonal mesostructured AIPO reported by us had a less condensed framework, so that a microporous material formed after calcination in air.⁴ Based on this result, it is expected that mesoporous AIPOs can be obtained by utilizing mesostructured AIPOs with larger hexagonal arrays. Here, we report the successful synthesis of thermally stable mesoporous AIPOs, based on this strategy.

Hexagonal mesostructured AIPOs were prepared by using C₁₆H₃₃NMe₃Cl (C₁₆TMACl) and C₂₂H₄₅NMe₃Cl (C₂₂TMACl), denoted as C₁₆-AIPO and C₂₂-AIPO, respectively. C₁₆-AIPO was prepared according to our previous paper.⁴ C₂₂-AIPO was prepared as follows: C₂₂TMACl (containing a small amount of shorter alkyl chains), tetramethylammonium hydroxide (TMAOH, 25 mass% in water), 85% H₃PO₄, and water were mixed for several hours. Aluminium triisopropoxide was added to this mixture under vigorous stirring, and the stirring was continued for 1 day. The composition of the starting mixture was Al₂O₃:P₂O₅:C₂₂TMACl:2.0 TMAOH:65.0 H₂O. The starting mixture was dispersed in distilled water at 70 °C, with formation of a white solid. This solid was washed with distilled water heated at 70 °C repeatedly, and dried at 80 °C. Solubilization of 1,3,5-triisopropylbenzene (TIPBz) into surfactant assemblage was also carried out for C₂₂TMACl; TIPBz has already been known to act as a solubilizing agent.⁹ The synthesis procedure was the same as described above and TIPBz was added before the addition of

aluminium triisopropoxide. The molar ratio of TIPBz/C₂₂TMACl in the starting mixture was 1, and this product was denoted as TIPBz/C₂₂-AIPO; the molar ratio of TIPBz/C₂₂TMACl in the TIPBz/C₂₂-AIPO was *ca.* 0.67 based on the CHN analysis, indicating that not all the TIPBz was solubilized in the surfactants. All the products were heated at 600 °C for 1 h in flowing N₂, followed by calcination at 600 °C for 1 h in flowing air. The calcined products were analyzed by XRD, N₂ adsorption, and ²⁷Al and ³¹P MAS NMR spectroscopy performed on a JEOL GSX-400 with a spinning rate of 5 kHz (for ²⁷Al: frequency: 104.05 MHz; pulse angle: 45°; recycle time: 5 s, for ³¹P: 161.70 MHz; 60°; 20 s, respectively).

The XRD patterns of all the products before calcination are shown in Fig. 1. The XRD peaks of the C₁₆-AIPO can be assigned to (100), (110) and (200) of a hexagonal phase. The XRD peaks of both C₂₂-AIPO and TIPBz/C₂₂-AIPO can also be assigned to (100), (110), (200) and (210) of hexagonal phases. The lattice parameters (*a*₀) of the C₁₆-AIPO, C₂₂-AIPO and TIPBz/C₂₂-AIPO were 4.8, 5.8 and 7.2 nm, respectively. These XRD results indicate the formation of mesostructured AIPOs with large hexagonal arrays by utilizing C₂₂TMA and solubilization of TIPBz.

In the XRD patterns of the calcined products (Fig.1), the *d*₁₀₀ peak shifted from 4.1 to 2.8 nm for C₁₆-AIPO after calcination, and the XRD patterns showed a shift from 5.1 to 3.7 nm for C₂₂-AIPO and a shift from 6.2 to 4.6 nm for TIPBz/C₂₂-AIPO. The lattice parameters (*a*₀: 2*d*₁₀₀/√3) of the calcined C₁₆-AIPO, calcined C₂₂-AIPO and calcined TIPBz/C₂₂-AIPO were 3.2, 4.3 and 5.3 nm, respectively, if we assume that the calcined products retain the hexagonal structures. The shrinkages are

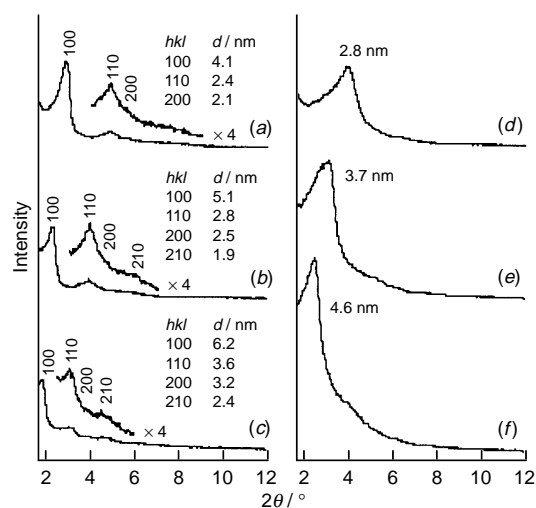


Fig. 1 XRD patterns of (a) C₁₆-AIPO, (b) C₂₂-AIPO and (c) TIPBz/C₂₂-AIPO, (d) calcined C₁₆-AIPO, (e) calcined C₂₂-AIPO and (f) calcined TIPBz/C₂₂-AIPO. These patterns were recorded on a Mac Science M03XHF²² diffractometer with monochromated Fe-K α radiation.

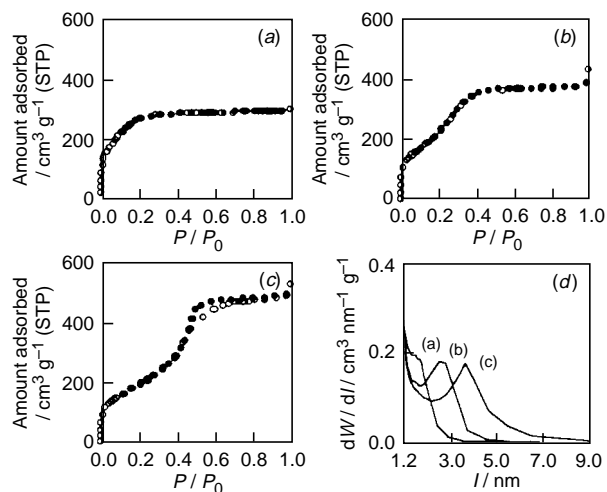


Fig. 2 N₂ adsorption isotherms of calcined (a) C₁₆-AIPO, (b) C₂₂-AIPO and (c) TIPBz/C₂₂-AIPO. Filled symbols denote desorption. Corresponding pore size distributions of the calcined products are shown in (d). These isotherms were obtained by a BELSORP 28 (Bel Japan, Inc.) at 77 K.

larger than those observed for silica-based mesoporous materials² because of less condensed frameworks as described below. In addition, the XRD peaks of all the products were broadened after calcination. These results suggest that the porous structures remained, although the sizes and the regularities of the hexagonal arrays decreased upon calcination.

Hexagonal arrays with relatively ordered arrangements were observed in the TEM images of all the products before calcination. After calcination, disordered arrays were obtained, and the periodic distances of these arrays decreased. These TEM images are in agreement with the XRD results. No other phases were detected in any TEM images.

The N₂ adsorption isotherms and the pore size distributions of the calcined products, calculated by the Horváth-Kawazoe method,¹⁰ are shown in Fig. 2. The N₂ adsorption isotherm of calcined C₁₆-AIPO showed almost type I behavior, indicating the presence of micropores; the pore diameter was *ca.* 1.8 nm. In contrast, the isotherms of the calcined C₂₂-AIPO and TIPBz/C₂₂-AIPO showed type IV behavior. The calcined C₂₂-AIPO had a BET surface area of 760 m² g⁻¹, a pore volume of 0.56 cm³ g⁻¹ and an average pore diameter of 2.8 nm. For the calcined TIPBz/C₂₂-AIPO, the BET surface area and the pore volume were 720 m² g⁻¹ and 0.72 cm³ g⁻¹, respectively. The pore diameter was *ca.* 3.9 nm.

The structural changes during calcination can be monitored by ²⁷Al and ³¹P MAS NMR measurements before and after calcination. The ³¹P MAS NMR spectra of the uncalcined products showed several peaks in the range δ 0 to -20, indicating the uncalcined products had less condensed frameworks.^{4,11,12} A broad peak centered at δ -21 was observed after calcination indicating condensation of the frameworks.^{11,12} The ²⁷Al MAS NMR spectra of the uncalcined products showed that signals due to both four- (Al^{IV}) and six-coordinate Al (Al^{VI}) at δ *ca.* 43 and 1, respectively. Based on the Al^{IV}/Al^{VI} intensity ratios, varying from *ca.* 0.6 (C₁₆-AIPO) to *ca.* 0.3 (C₂₂-AIPO and TIPBz/C₂₂-AIPO), a large amount of Al^{VI} is present in the uncalcined products and they may be coordinated with not only PO₄ units but also water molecules,⁴ being related to less condensed frameworks. Only a broad signal due to Al^{IV} was observed at *ca.* δ 39 for each calcined product; the chemical shift strongly suggesting that the Al^{IV}VO₄ units are surrounded by mainly P.^{11,13} Although the Al/P ratios (1.5), discussed below, might suggest the presence of Al-O-Al bonds, such

bonds are unlikely, judging from very different chemical shifts expected for [Al^{IV}VO₄(nAl)] (*n* = 1-4).¹³

Both N₂ adsorption and NMR results indicate that the pore sizes can be controlled although shrinkage must be taken into consideration, so that the C₂₂-AIPO and TIPBz/C₂₂-AIPO changed to mesoporous materials after calcination owing to the larger hexagonal arrays.

The C_nTMA/(Al + P) ratios were 0.25 (C₁₆-AIPO), 0.19 (C₂₂-AIPO) and 0.16 (TIPBz/C₂₂-AIPO), and the Al/P ratios of all the products were *ca.* 1.5. This Al/P value is different from unity which would be expected for an ideal three-dimensional AlPO₄-*n*.³ The Al/P ratios of mesostructured and/or mesoporous aluminophosphates reported by Feng *et al.*,⁵ Zhao *et al.*,⁶ Holland *et al.*,⁷ and Cheng *et al.*⁸ are 2.9, 0.64, 1.5, and 1.0, respectively. Although the Al/P ratios of mesoporous AIPOs reported here are similar to those reported by Holland *et al.*,⁷ the coordination of Al atoms is six for their materials while it is mainly four for ours. For the other three materials, the Al/P ratios are clearly different. Nevertheless, the coordination of Al atoms in the material reported by Feng *et al.*⁵ is similar to ours. However, because there is little information on the aluminophosphate frameworks in those reports, appropriate comparison with our materials is not possible at present.

In conclusion, thermally stable mesoporous AIPOs were obtained by calcination of hexagonal mesostructured AIPOs prepared by using both C₂₂TMA as a micelle aggregate and TIPBz as a solubilizer. This method can be applicable for the preparation of various mesoporous materials from less condensed inorganic units-surfactant mesostructured materials, which normally afford microporous materials owing to substantial shrinkage. A study on the surface structure of these mesoporous AIPO is now in progress.

The authors acknowledge Mr M. Fuziwara, MCCL, Waseda University for TEM measurements. K. K. acknowledges the financial assistance from Grant-in-Aid for the Special Priority Area by the Ministry of Education, Science, and Culture of the Japanese Government.

Notes and References

*E-mail: kuroda@mn.waseda.ac.jp

- 1 T. Yanagisawa, T. Shimizu, K. Kuroda and C. Kato, *Bull. Chem. Soc. Jpn.*, 1990, **63**, 988; S. Inagaki, A. Koiwai, N. Suzuki, Y. Fukushima and K. Kuroda, *Bull. Chem. Soc. Jpn.*, 1996, **69**, 1449.
- 2 C. T. Kresge, M. E. Leonowicz, W. J. Roth, J. C. Vartuli and J. S. Beck, *Nature*, 1992, **359**, 710.
- 3 S. T. Wilson, B. M. Lok, C. A. Messina, T. R. Cannan and E. M. Flanigen, *J. Am. Chem. Soc.*, 1982, **104**, 1146.
- 4 T. Kimura, Y. Sugahara and K. Kuroda, *Book of Abstracts, 11th Int. Zeolite Conf.*, Seoul, Korea, 1996, RP45; T. Kimura, Y. Sugahara and K. Kuroda, *Chem. Lett.*, 1997, 983.
- 5 P. Feng, Y. Xia, J. Feng, X. Bu and G. D. Stucky, *Chem. Commun.*, 1997, 949.
- 6 D. Zhao, Z. Luan and L. Kevan, *Chem. Commun.*, 1997, 1009; D. Zhao, Z. Luan and L. Kevan, *J. Phys. Chem.*, 1997, **101**, 6943.
- 7 B. T. Holland, P. K. Isbester, C. F. Blanford, E. J. Munson and A. Stein, *J. Am. Chem. Soc.*, 1997, **119**, 6796.
- 8 S. Cheng, J.-N. Tzeng and B.-Y. Hsu, *Chem. Mater.*, 1997, **9**, 1788.
- 9 A. Mochizuki, M. Yamai and S. Namba, *Book of Abstracts, 11th Int. Zeolite Conf.*, Seoul, Korea, 1996, RP29.
- 10 G. Horváth and K. Kawazoe, *J. Chem. Eng. Jpn.*, 1983, **16**, 470.
- 11 D. Müller, E. Jahn, G. Ladwig and U. Haubenreisser, *Chem. Phys. Lett.*, 1984, **109**, 33.
- 12 R. F. Mortlock, A. T. Bell and C. J. Radke, *J. Phys. Chem.*, 1993, **97**, 767.
- 13 D. Müller, W. Gessner, A. Samoson, E. Lippmaa and G. Scheler, *J. Chem. Soc., Dalton Trans.*, 1986, 1277.

Received in Cambridge, UK, 24th November 1997; 7/08463C